

Glyoxal cross-linking of solubilised extracellular matrix to produce highly porous, elastic and chondro-permissive scaffolds for orthopaedic tissue engineering

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ABSTRACT

Extracellular matrix (ECM) derived implants hold great promise for tissue repair, but new strategies are required to produce efficiently decellularized scaffolds with the necessary porosity and mechanical properties to facilitate regeneration. In this study we demonstrate that it is possible to produce highly porous, elastic, articular cartilage ECM derived scaffolds that are efficiently decellularized, nonimmunogenic and chondro-permissive. Pepsin solubilised porcine articular cartilage was cross-linked with glyoxal, lyophilized and then subjected to dehydrothermal treatment. The resulting scaffolds were predominantly collagenous in nature, with the majority of sGAG and DNA removed during scaffold fabrication. Four scaffold variants were produced to examine the effect of both ECM (10 or 20 mg/ml) and glyoxal (5mM or 10mM) concentration on the mechanical and biological properties of the resulting construct.. When seeded with human infrapatellar fat pad derived stromal cells (FPSCs), the scaffolds with the lowest concentration of both ECM and glyoxal were found to promote the development of a more hyaline-like cartilage tissue, as evident by increased sGAG and type II collagen deposition. Furthermore, when cultured in the presence of human macrophages, it was found that these ECM derived scaffolds did not induce the production of key pro-inflammatory cytokines, which is critical to success of an implantable biomaterial. Together these findings demonstrate that the novel combination of solubilised articular cartilage ECM and glyoxal crosslinking can be used to produce highly porous scaffolds that are sufficiently decellularized, highly elastic, chondro-permissive and do not illicit a detrimental immune response when cultured in the presence of human macrophages.

Keywords: Extracellular matrix, Cartilage, Scaffold, Decellularization, Tissue engineering, Immune response.

1. Introduction

Due in part to its lack of vasculature, articular cartilage possesses a relatively poor intrinsic capacity for repair and as such external factors such as cells, growth factors, biomaterials or a combination of these must be delivered to the site of injury to promote repair of damaged joints¹⁻³. Current clinical treatment options for articular cartilage repair such as micro-fracture remain sub-optimal, with the repair tissue generally consisting of bio-mechanically inferior fibro-cartilage⁴. Xenogenic extracellular matrix (ECM) derived biological scaffolds have been used in a variety of clinical indications, promoting tissue repair by providing both structural and functional cues to resident cells^{5,6}. ECM scaffolds fabricated from decellularized animal or human articular cartilage tissue are promising biomaterials for articular cartilage regeneration due to their inherent chondro-inductivity⁷⁻¹⁹. Such biological scaffolds typically take one of two forms. Particulated or solubilised cartilage ECM can be freeze dried and subsequently cross-linked, resulting in porous scaffolds with relatively poor mechanical properties when compared to native articular cartilage²⁰⁻²². Alternatively, intact cartilage explants can be decellularized without disrupting the overall collagen network, resulting in a graft with more desirable mechanical properties but with a porosity that limits cellular infiltration²³⁻²⁵

The success or failure of any implantable biomaterial is also invariably linked to the host immune response. The host innate and adaptive immune system can respond to cellular antigens and DNA within ECM derived constructs, or various components of the ECM itself. Ineffectively decellularized ECM scaffolds retain remnants of cellular material which can elicit proinflammatory cytokine production and adverse tissue remodelling²⁶⁻²⁸. One of the key criteria set as confirmation of successful decellularization is to obtain a DNA content of less than 50ng per mg of dry weight²⁹. However, overly harsh decellularization protocols can have a detrimental impact on the bioactivity and biomechanical properties of the resulting biomaterial. Protocols that employ the use of detergents such as Sodium Dodecyl Sulfate (SDS)

can dramatically reduce the proteoglycan content of the tissue, negatively impacting the mechanical properties of the resulting grafts. Furthermore, freeze-thaw cycles that are commonly employed in decellularization protocols can result in disorder of the collagen network due to ice crystal formation^{6,30}.

In the case of scaffolds generated by freeze-drying particulated or solubilized ECM, physical and chemical crosslinking methodologies are generally applied in order to maintain the structural integrity and porosity of the resulting scaffold architecture^{31,32}. However, the use of chemical crosslinkers can have a detrimental effect due to the cytotoxicity of the reagents used. Glutaraldehyde is a commonly used chemical crosslinker for collagen based biomaterials, however cytotoxicity can become an issue at higher concentrations^{33,34}. Biomaterials which have been cross-linked with glutaraldehyde can also induce a strong humoral immune response *in vivo*³⁵. In addition, chemical crosslinking can also attenuate ECM scaffold degradation *in vivo* and as such negatively alter the host response to the implanted biomaterial³⁶. Therefore, the degree and method of crosslinking *are* vital parameters to consider when using ECM-derived scaffolds for tissue repair applications.

This study examines the use of glyoxal as a chemical crosslinker, in addition to physical crosslinking using dehydrothermal (DHT) treatment, to produce porous and highly elastic ECM derived scaffolds for musculoskeletal tissue engineering. The crosslinking chemistry of glyoxal is similar to that of glutaraldehyde but with a reduced cytotoxic profile³⁷. By solubilizing articular cartilage ECM prior to cross-linking and freeze-drying, we hypothesize that it is possible to produce efficiently decellularized, yet mechanically competent, ECM derived scaffolds that are nonimmunogenic and capable of facilitating the differentiation of adult stem cells towards a hyaline cartilage phenotype. Our specific goals were to fabricate and ascertain the biochemical and physical characteristics of the resulting scaffolds and the extent of decellularization. To establish the optimal conditions for chondrogenic differentiation

of adult stem cells, we varied both the concentration of ECM in the scaffolds and the degree of chemical crosslinking. Finally, we examined the effect of the implant decellularization process on cytokine production by macrophages, key cellular regulators of tissue repair and inflammation.

2. Materials and Methods

2.1 Articular cartilage ECM solubilisation.

Articular cartilage (AC) was harvested, under sterile conditions, from the femoral condyle and trochlear ridge of 3-4 month old female pigs. The articular cartilage was shaved from the joint using a biopsy punch and subsequently diced into 1-2mm pieces using a scalpel. The cartilage pieces were pre-treated with 0.2M NaOH solution for 24 hours at 4°C with gentle agitation. Pre-treatment extracts the majority of sGAGs in the tissue as well as removing cellular components³⁸. After removal of the pre-treatment solution the tissue was washed in sterile dH₂O. The cartilage was subsequently solubilised using a solution of 1500U/ml of pepsin (Sigma) in 0.5M acetic acid (Sigma) under rotation(4RPM) for 24 hours at room temperature. To remove insoluble material the tissue sample was centrifuged at 2500G for 1 hour. The supernatant containing the solubilised tissue was next transferred to a new tube and combined with a 5M NaCl (Sigma) solution to a final concentration of 0.9M NaCl in order to preferentially salt precipitate type II collagen from the sample³⁹. The salt and collagen solution were mixed followed by equilibration overnight at 4°C and subsequently centrifuged at 2500G for 1 hour at 4°C. The supernatant was discarded, and the collagen pellet was resuspended in 0.5M acetic acid followed by rotation at 4RPM overnight at room temperature to fully solubilise the collagen. The salt precipitation procedure was then repeated a second time. The acid solubilised collagen was then transferred into a dialysis membrane (MWCO 6-8 kD

Spectrum Labs). Dialysis was performed against 0.02M Na₂HPO₄ for 48 hours at 4°C with agitation of the dialysate. The dialyzing solution was changed for fresh solution after 24 hours. Following dialysis, the solubilised AC solution was transferred to 24 well tissue culture plates (1ml/well) and lyophilized. Once lyophilized the solubilised AC was stored at -80°C prior to scaffold fabrication.

2.2 Scaffold fabrication

Four variants of AC-ECM scaffolds were fabricated, with two ECM concentrations and two cross-linker concentrations: 20mg/ml AC crosslinked with 10mM glyoxal (20mg/10mM), 20mg/ml AC crosslinked with 5mM glyoxal (20mg/5mM), 10mg/ml AC crosslinked with 10mM glyoxal (10mg/10mM) and 10mg/ml AC crosslinked with 5mM glyoxal (10mg/5mM). The concentrations of ECM and glyoxal were chosen based on preliminary optimization experiments (data not shown) which examined cell infiltration and glyoxal toxicity. Glyoxal was chosen as a cross-linker due to the fact that it results in relatively low levels of cytotoxicity when compared to other aldehyde cross-linkers such as glutaraldehyde⁴⁰. In order to fabricate AC-ECM scaffolds, the lyophilized solubilised AC was first weighed using a balance and transferred to sterile tubes. Under sterile conditions, the lyophilized AC was resuspended in high glucose Dulbecco's Modified Eagle Medium (DMEM - Gibco) to yield a final concentration of 20mg/ml or 10mg/ml. The phenol red in the DMEM was used as an indicator of pH. To dissolve the AC, 80µl of 0.5M Acetic acid was added per ml of AC-ECM solution to yield a final concentration of 0.02M Acetic acid, the solution was pipetted several times to ensure homogeneous mixing. The AC solution was neutralised by adding 20µl volumes of 0.1M NaOH until the colour of the solution changed from yellow to red. The AC solution was then chemically crosslinked using glyoxal (Sigma) at a final concentration of 10mM or 5mM.

The solution was mixed well and incubated for 30 mins at 37°C to allow crosslinking to occur. After crosslinking the solution was transferred to custom made moulds (height 5mm and diameter 3mm) and freeze dried (FreeZone Triad, Labconco, KC, USA). To create porous scaffolds, the solution was initially frozen to -30°C at a rate of -1°C/min, this temperature was held for 1 hour before rising to -10°C at a rate of +1°C/min. Once at -10°C, this temperature was maintained under a vacuum of 0.2 mBar for 24 hours before the temperature was increased to +20°C at a rate of +1°C/min⁹. Scaffolds were physically crosslinked in a vacuum oven (VD23, Binder, Germany) by dehydrothermal (DHT) treatment at 115°C for 24 hours under vacuum (2 mBar).

To act as a positive control for immunological experiments, scaffolds were also fabricated from non-solubilised, particulated porcine articular cartilage as previously described¹⁰. Briefly, articular cartilage was harvested and then cryomilled (6770 Freezer/Mill, SPEX, UK) to a fine powder. This powder was then resuspended in dH₂O to create a 250mg/ml slurry. The slurry was then transferred into 5mm by 3mm moulds and was freeze-dried and DHT crosslinked using the same parameters as described above.

2.3 Scanning Electron Microscopy (SEM), pore size and porosity determination

Prior to SEM, scaffolds were sputter coated (108 auto sputter coater, Cressington, UK) with a gold/palladium alloy (Agar Scientific, UK) for 90 seconds at 0.1mBar. SEM images were obtained using a Zeiss Ultra Plus (Zeiss, Germany) with an acceleration voltage of 5kV and working distance of 5mm. To quantify the mean pore size of the various scaffolds, 5 images (containing a minimum of 100 pores) from 3 different scaffolds were measured using the analyse particle function of imageJ (National Institutes of Health Bethesda, Maryland) following thresholding of the images. Scaffold porosity was determined by gravimetry as

previously described^{41,42}. Scaffolds were weighed using a digital mass balance and dimensions obtained using a vernier callipers in order to calculate density values.

2.4 Fat pad stromal cell (FPSC) isolation and expansion

Ethical approval for the donation of human infrapatellar fat pad (IFP) tissue and subsequent isolation of fat pad stromal cells (FPSCs) from the tissue was obtained from the institutional review board of the Mater Misericordiae University Hospital Dublin. IFP tissue was obtained from three patients undergoing anterior cruciate ligament (ACL) reconstruction surgery. All three donors were male with a mean age of 28.6 years (26, 26 and 34 years old). FPSCs were isolated from IFP tissue as previously described⁴³⁻⁴⁸. Briefly, IFP tissue was washed with Phosphate Buffered Saline (PBS Sigma) followed by manual dicing with scalpels. IFP tissue was then incubated in high glucose DMEM (Gibco) containing 750U/ml collagenase II (Worthington Biochemical, NJ, USA) with Penicillin (100 U/ml) and streptomycin (100µg/ml) (both Gibco) for 3-4 hours until the tissue was fully digested. After digestion, insoluble tissue and residual fat was removed and discarded. The remaining cells were passed through cell strainers (Fisher Scientific), centrifuged and washed with D-PBS. Cells were then plated in T-175 flasks (Fisher Scientific) at a seeding density of 5×10^3 cells/cm². FPSCs were expanded in DMEM supplemented with 10% FBS (Gibco), with Penicillin (100 U/ml) and streptomycin (100µg/ml) (both Gibco) and 5ng/ml fibroblast growth factor-2 (FGF-2, ProSpec-Tany, Israel) as previously described⁴⁹. Expansion media was changed three times per week. FPSCs were not used beyond passage 3 for subsequent differentiation experiments.

2.5 Seeding AC-ECM scaffolds

Initially, to remove any residual DMEM all scaffolds were washed and rehydrated in D-PBS. To enhance seeding efficiency, the scaffolds were placed inside agarose moulds with a diameter of 5mm. The agarose moulds containing the scaffolds were then placed into 24-well

tissue culture plates (Fisher Scientific). To examine the ability of the various AC scaffolds to promote the chondrogenic differentiation of FPSCs *in vitro*, 0.5×10^6 FPSCs were seeded onto individual scaffolds suspended in 25 μ l of expansion media. FPSCs were allowed to attach to the scaffolds for 1 hour in an incubator at 37°C. After FPSC attachment, 2.5ml of chemically defined chondrogenic differentiation media (CDM) was added per well. CDM consisted of high glucose DMEM supplemented with Penicillin (100 U/ml) and streptomycin (100 μ g/ml) (both Gibco), 100 μ g/ml sodium pyruvate (Sigma), 40 μ g/ml L-Proline (Sigma), 50 μ g/ml L-ascorbic acid-2-phosphate (Sigma), 1.5mg/ml bovine serum albumin (BSA-Sigma), 1X insulin transferrin selenium (ITS- Gibco), 100nM dexamethasone (Sigma) and 10ng/ml transforming growth factor beta-3 (TGF- β 3 – ProSpec Tany, Israel). The FPSC seeded constructs were maintained in CDM for 28 days with the CDM being replenished three times per week.

2.6 Biochemical analysis of constructs

Cell free scaffolds (Day 0) and FPSC seeded constructs (Day 28) were analysed for DNA and sGAG content. Prior to performing assays, scaffolds and constructs were enzymatically digested with papain (125 μ g/ml - Sigma) in a buffer containing 100mM Sodium Phosphate (Sigma) with 5mM Na₂EDTA (Sigma) at pH 6.5 as previously described⁴³. sGAG quantification was performed using a 1, 9 dimethylmethylene blue (DMMB) assay according to the manufacturer's protocol with bovine tracheal chondroitin 4- sulfate used as a reference standard (Blyscan sulfated sGAG assay kit, Biocolor, Northern Ireland). Quantification of dsDNA in the digested constructs was performed using a Quant-iT Pico Green dsDNA kit (Invitrogen) according to the manufacturer's protocol. The combination of results from the DMMB and pico-green assays provides a ratio of sGAG normalized to dsDNA content. Collagen content of scaffolds was quantified by measuring total hydroxyproline content as

previously described⁹. A hydroxyproline:collagen ratio of 1:7.69 was assumed to determine the collagen content⁵⁰.

2.7 Assessment of endotoxin contamination

ECM scaffolds were first tested for lipopolysaccharide (LPS) contamination using the HEK-Blue™ hTLR4 assay system (Invivogen). HEK-blue cells (5×10^5 cells/ml) expressing TLR4 were stimulated with LPS (1–100 ng/ml; positive control), or ECM scaffolds for 24 h. The expression of SEAP which is under the control of NF- κ B and AP-1 was tested by incubating cell supernatants with HEK-blue detection medium for 30 min at 37 °C and absorbance was read at 650 nm.

2.8 Analysis of cytokine production by macrophages upon seeding onto AC ECM derived scaffolds

Peripheral blood mononuclear cells (PBMC) were isolated by means of density gradient centrifugation from leukocyte-enriched buffy coats from anonymous healthy donors, obtained with permission from the Irish Blood Transfusion Board, St. James's Hospital, Dublin. CD14⁺ myeloid cells were purified from PBMC by means of positive selection using anti-CD14 magnetic beads according to the manufacturer's protocol (Miltenyi Biotec, Bergisch Gladbach, Germany). Cells were shown to be >90% pure as determined by flow cytometry. CD14⁺ monocytes were differentiated into macrophages for 6 days in the presence and absence of ECM derived scaffolds using M-CSF (50 ng/ml). As a positive control, macrophages were treated with 100 ng/ml Lipopolysaccharides (LPS) for 24 hours. IL-6, IL-8, TNF α and IL-10 cytokine production was quantified in cell supernatants by ELISA (eBiosciences).

2.9 Mechanical testing

Scaffolds ($\Phi 5$ mm x h 3 mm, n = 4) were subjected to a uniaxial unconfined compression test in PBS using a single column mechanical tester (Zwick/Roell Z2.5, Herefordshire, UK) with a 5N load cell. Briefly, constructs were kept hydrated in PBS bath maintained at room temperature. A preload of 0.01N was applied for 60 seconds to ensure that top and bottom construct surfaces were in direct contact with the impermeable loading platens. Force was zeroed after preload, then followed by a cyclic compression test consisting of 6 compressive cycles with increasing strain amplitude of 10, 20, 30, 40, 50 and 60% in sequence at a rate of 2.22% strain per second. A holding time of 10 seconds between each cycle was added to allow full height recovery of the samples. The load versus displacement data were recorded throughout. The apparent stress and strain were calculated by dividing the load value with the initial cross-sectional area of each sample and the displacement value with the initial sample height, respectively. Cumulative percentage of the total height loss and height loss per cycle was calculated to characterize the super-elastic properties of the constructs. Scaffold permanent deformation (PD) at each cycle was calculated as follows: $PD = (\text{Test Speed} * \Delta t_n) / h_0$, where Δt_n is the interval of time at the start of the n^{th} cycle in which no force is applied to the sample, while h_0 is the initial height of the sample. Samples were also subjected to 80% strain during 10 cycles at 2.22% strain per second (supplementary data). Differences between Young's modulus between 1st, 5th and 10th cycle were compared to evaluate elasticity. Data was normalised to 1st cycle in each group. To obtain the equilibrium modulus of cell-free scaffolds, stress relaxation tests were performed whereby 30% strain was applied followed by relaxation until equilibrium was achieved. For cell seeded constructs, 10% strain was applied during stress relaxation tests to obtain the equilibrium modulus. Immediately after this dynamic tests were performed by applying a cyclic strain of 1% at 1 Hz to determine the dynamic modulus of the engineered constructs.

2.10 Histology and immunohistochemistry

All samples were washed in PBS followed by overnight fixation in 4% paraformaldehyde (Sigma, Ireland). Samples were dehydrated and wax embedded. Embedded constructs were then sectioned at a thickness of 5µm using a microtome. Sections were stained with 1% alcian blue 8X in 0.1M HCl (Sigma, Ireland) to examine sulphated glycosaminoglycan (sGAG) and picrosirius red to examine collagen deposition. To identify the specific collagen types present in the constructs, immunohistochemistry was performed for collagen type I, type II or X as previously described ⁴⁷. Briefly, after dewaxing and rehydrating the sections antigen retrieval was performed by incubation with Chondroitinase ABC for collagen types I and II or with Pronase for collagen type X. After blocking for non-specific binding, sections were incubated with primary antibody (anti collagen type I (1:400), type II (1:100) or type X (1:200) all HRP-conjugated, all Abcam, UK) overnight at 4°C. Endogenous peroxidase activity was blocked with hydrogen peroxide (Sigma) prior to incubation with the anti-mouse IgG secondary antibody (Sigma). Sections were then incubated with 3,3'-diaminobenzidine (*DAB*) peroxidase substrate (Vector Labs, UK) to visualise positive staining.

2.11 Statistical analysis

All experiments were performed on three separate human donors. Results are presented as mean +/- standard deviation. Statistical analysis was performed using Graph Pad Prism (San Diego, USA). Statistical differences were analysed by one-way analysis of variance (ANOVA) with Tukey's test for multiple comparisons to compare experimental conditions. Statistically significant changes are marked as * = $p < 0.05$; ** = $p \leq 0.01$; *** = $p \leq 0.001$

3. Results

3.1 ECM scaffold porosity, composition and mechanical properties

Highly elastic and porous scaffolds can be fabricated from pepsin solubilised cartilage ECM which have been cross-linked with glyoxal prior to freeze drying (Fig. 1). All scaffolds were found to display highly elastic mechanical properties when manually compressed (Fig. 1A), returning to their original shape after unloading. Histological and biochemical analysis of the solubilised ECM derived scaffolds demonstrated almost complete removal of sGAG from the matrix (Fig. 1 B&E), with strong staining for picrosirius red confirming the collagenous nature of the scaffolds. Collagen content was quantified by hydroxyproline assay, between 49% and 24% of the native dry weight of the collagen was maintained post scaffold fabrication (Fig. 1G). Scanning electron micrographs (SEM) of the scaffolds revealed a uniform pore architecture with the majority of pores having a diameter between 15-20 μ m (Fig. 1 B&C). All scaffolds were found to have a porosity between 97% and 98% (Fig. 1D). The amount of xenogeneic DNA remaining in the scaffolds post processing was also quantified. The mean quantity of DNA observed across all scaffolds tested was 39 ± 1.5 ng/mg of ECM dry weight (Fig. 1F).

Having fabricated four variations of highly elastic and porous scaffolds that had similar physical characteristics to each other, we then sought to mechanically evaluate the various scaffolds to ascertain if varying the concentrations of ECM and cross-linker can affect the mechanical properties of the resulting scaffolds. Mechanical testing of the scaffolds revealed that the concentration of cross-linker had no discernible effect on the mechanical properties of the scaffolds. Increasing the concentration of ECM in the scaffolds from 10mg/ml to 20mg/ml significantly increased the equilibrium modulus. These scaffolds were also stiffer than those generated using traditional strategies of freeze-drying particulated articular cartilage ECM

slurry (AC-Cryomill) (Figure. 2A). To further evaluate the elastic properties of the scaffolds, a compression test consisting of 6 compressive cycles with increasing strain amplitude of 10, 20, 30, 40, 50 and 60% in succession was performed. A relatively elastic stress-strain response was observed at all amplitudes examined, with a small degree of hysteresis observed (Figure. 2B). Relatively minor levels of residual deformation were detected at the end of each loading cycle, which did accumulate with each loading cycle (Figure. 2B). Additional mechanical tests were also undertaken to demonstrate the different behaviour of solubilised ECM scaffolds and those produced using particulated ECM (AC-Cryomill). The solubilised ECM scaffolds were shown to undergo lower levels of permanent deformation after being compressed by 20% strain (Supp Fig.2A) and unlike the particulated ECM scaffolds, their stiffness did not reduce upon the application of high magnitude cyclic stain (Supp Fig.2B).

3.2 Chondrogenesis within solubilised ECM derived scaffolds

Having ascertained the physical, biochemical and mechanical properties of the scaffold variations, we next evaluated their chondrogenic-capacity by seeding them with FPSCs and maintaining the constructs in culture with TGF- β 3. After 28 days in culture, all FPSC seeded scaffolds appeared cartilage-like in appearance, exhibiting a glossy structure (Fig. 3B). (Fig. 3A). All engineered tissues stained positive for alcian blue and picrosirius red, indicative of cartilage matrix synthesis. Similar histological staining of cell-free scaffolds is provided in Figure 1A. The most robust staining for new ECM deposition was observed for scaffolds fabricated using lower ECM (10mg) and lower glyoxal cross-linking (5mM) concentrations (Fig. 3B). Quantitative biochemical analysis substantiated the results observed histologically, with the 10mg/5mM scaffold supporting the highest levels of sGAG accumulation (GAG:DNA) of all groups examined (Fig. 3C-E). To confirm the development of cartilage-specific matrix within the ECM derived scaffolds, we next analysed the deposition of collagen

type I, II and X within the engineered tissues by immunohistochemistry (Figure 4). Type II collagen accumulation was observed in the 20mg/5mM, 10mg/10mM and 10mg/5mM groups, with little positive staining observed in the 20mg/10mM group. All engineered constructs stained weakly for collagen types I and X. To assess the mechanical properties of the engineered tissues after 28 days in culture, the constructs were subjected to unconfined compression testing to 10% strain. As expected the 10mg/5mM scaffold group, which supported the greatest levels of sGAG and type II collagen synthesis, were the stiffest constructs in terms of both equilibrium and dynamic modulus (Fig. 5 A&B)

3.3 Macrophage response to solubilised ECM

Having established that the 10mg/5mM scaffold best supported chondrogenesis of FPSCs, we next sought to evaluate the immune response to these optimized scaffolds. To first establish that the scaffold preparations were endotoxin (LPS) free, we tested the ECM scaffolds for LPS contamination and TLR4 activation as determined using the HEK-Blue LPS detection kit. We found that the concentration of endotoxin (LPS) was below the threshold level for TLR4 activation upon treatment with both native cartilage and solubilised AC ECM (Figure S1)

Immunogenicity of the native scaffolds was next assessed by culturing primary human macrophages in the presence of native cartilage or solubilised AC ECM. As a positive control, cells were stimulated with LPS to drive pro-inflammatory cytokine production. While particulated (cryomilled) articular cartilage promoted high levels of IL-6 production, this was reduced to control levels in the solubilised AC scaffold group (Fig. 6A) and there was a trend (albeit not significant) towards increased TNF α and IL-10 production (Fig. 6B & D). In contrast, no such increases were observed when macrophages were cultured with solubilised AC ECM. Interestingly, production of the chemokine, IL-8, was higher in the solubilized ECM scaffold versus particulated AC groups (Fig. 6C).

4. Discussion

The overall goal of this study was to fabricate a porous articular cartilage ECM derived scaffold that (i) was efficiently decellularized, (ii) was highly elastic with shape memory properties, (iii) facilitated the chondrogenic differentiation of adult stem cells and (iv) did not illicit a negative immune response when in contact with human macrophages. We developed four different scaffolds, with variations in the concentrations of ECM and degree of cross-linking, all of which were found to be sufficiently decellularized. The concentrations of ECM and glyoxal were chosen based on preliminary basic optimization experiments (data not shown) which examined cell infiltration and glyoxal toxicity. Glyoxal was chosen as a cross-linker due to the fact that it results in relatively low levels of cytotoxicity when compared to other aldehyde cross-linkers such as glutaraldehyde ⁴⁰. Scaffolds fabricated with 10mg/ml of AC-ECM crosslinked with 5mM glyoxal were found to promote the greatest deposition of sGAG and type II collagen by FPSCs. When cultured in the presence of macrophages, these scaffolds elicited an immune response believed more compatible with tissue regeneration (lower levels of the inflammatory cytokines IL-6 and TNF) when compared to non-decellularized, cryomilled AC scaffolds. However, we do note that IL-10 cytokine secretion was also decreased compared to native tissue. While this cytokine is considered an anti-inflammatory cytokine, and may be beneficial for the tissue regenerative process, the solubilisation protocol would appear to result in a reduction of global cytokine production rather than specifically reducing pro-inflammatory cytokines. Taken together, these results indicate that efficiently decellularized ECM derived scaffolds can be fabricated from solubilised porcine articular cartilage, and that these biomaterials are chondro-permissive.

Glyoxal cross-linking of the solubilised ECM produced elastic scaffolds exhibiting shape-memory properties. The solubilization process removed the majority of sGAGs from the

ECM, leaving behind a collagen rich solution. During the scaffold fabrication process, this solution is brought to neutral pH and 37°C, leading to collagen fibril formation and gelation. Exposure to the dialdehyde glyoxal subsequently cross-links this physical network by crosslinking free amine groups in the collagen^{37,51}. Freeze drying then produces a porous network of collagen struts displaying elastic properties at the macroscale. Previous mechanical tests on individual collagen fibrils has demonstrated that while they do not behave as homogenous materials, they commonly display linear behaviour, and can withstand tensile strains of up to 100% without fracturing⁵². The fibrils also exhibited time-dependent recoverable residual strains. Hence the elastic behaviour of these ECM derived scaffolds can be expected based on what is known about the mechanical properties of collagen itself. All scaffold variations demonstrated high elasticity when wet with shape-memory characteristics. In their current iteration the scaffolds should be capable of sustaining mechanically challenging environments due to their highly elastic properties, but will not initially contribute to load bearing until sufficient *de novo* tissue has formed. In order to improve mechanical properties of these scaffolds, different chemical crosslinking techniques could be employed such as EDAC/NHS crosslinking⁹ or naturally derived crosslinking agents, such as genipin⁵³. Additionally, the scaffolds could be reinforced with a polymeric (e.g. polycaprolactone) network or frame to provide additional mechanical support;⁵⁴ while this would increase the compressive modulus of the construct a reduction in elasticity would also result. Alternatively, the solubilized AC-ECM could be engineered into a methacrylated hydrogel which can be subsequently photo-crosslinked to create a construct that has previously been demonstrated to possess a compressive modulus approaching the range of native articular cartilage¹².

Having successfully fabricated the four scaffold variants, we then analysed the levels of xenogeneic DNA remaining in the scaffolds; 39 (+/-1.46) ng DNA per mg of dry weight was found to be the mean residual DNA level across the four variants. The AC solubilisation

procedure successfully removed ~95% of xenogeneic DNA from the biomaterial. This suggests that the resulting scaffolds are sufficiently decellularized for clinical use, based on established recommendations of <50 ng of DNA per mg of (dry weight) tissue ²⁹. Along with xenogenic DNA removal, the ultimate test to fully evaluate the effectiveness of decellularization protocol would be to employ the use of an *in vivo* model to assess the immune system response to the implant. Successful integration of ECM scaffolds is dependent on eliciting an appropriate host immune response ⁵⁵. Therefore preliminary assays were carried out to examine how primary human macrophages responded to these scaffolds. TNF α and IL-6 are potent pro-inflammatory cytokines produced by macrophages upon exposure to DNA or cellular antigens and were therefore examined as indicators of efficient decellularization. Non decellularized, cryomilled AC derived scaffolds enhanced production of both cytokines, however this was not observed in the solubilised AC derived scaffolds groups, suggesting that the solubilisation process was successful in removing most cellular antigens that would usually elicit a detrimental immune response. These cytokines, in addition to IL-10, are also normally enhanced in the presence of endotoxin, which can often contaminate biological scaffolds⁵⁶ The absence of any of these cytokines in the solubilised AC scaffold groups also indicates that scaffolds are produced free of any major endotoxin contamination. Furthermore, TNF α and IL-6 have been shown to inhibit chondrogenic differentiation of MSCs ⁵⁷⁻⁵⁹, therefore the decreased production of these cytokines provides further evidence that the solubilised AC scaffold will provide an environment supportive of chondrogenesis if implanted *in vivo*.

Interestingly, high levels of the chemokine IL-8 were produced by macrophages cultured in the presence of solubilised AC scaffolds compared to the non-decellularized, cryomilled AC scaffolds. The precise role of this chemokine in chondrogenesis has not yet been established, however IL-8 in the presence of bone marrow concentrate has been shown to upregulate the

chondrogenic markers Sox9, ACAN and Col II in MSCs ⁶⁰ . It will be interesting to determine the precise role played by this chemokine in host interactions with ECM scaffolds. Of note, it has been shown that IL-8 can also promote osteoclastogenesis ⁶¹, therefore a certain level of this protein may actually be beneficial for optimal remodelling if such biomaterials were used for osteochondral defect repair.

Solubilised cartilage ECM derived scaffolds were found to facilitate robust chondrogenic differentiation of FPSCs. Adult stem cells have been isolated from several tissues including the bone marrow, ⁶²⁻⁶⁴ subcutaneous adipose tissue ^{65,66} and the infrapatellar fat pad (IFP) ^{8-10,45,47,67}. MSCs from these tissues have been demonstrated to readily differentiate into cartilage tissue and as such have been proposed as an ideal cell source for cartilage repair applications ⁶⁸. While less studied than bone marrow-derived MSCs, both the anatomical location of the IFP and the high proportion of progenitor cells found in the tissue make the IFP an ideal cell source for cartilage repair applications. Infrapatellar fat pad-derived stromal cells (FPSCs) have been demonstrated to have at least comparable chondrogenic capacity to bone marrow derived-MSCs ⁶⁹⁻⁷¹. Scaffolds fabricated with 10mg/ml of AC-ECM crosslinked with 5mM glyoxal were found to promote the greatest deposition type II collagen by FPSCs within the constructs. Type II collagen is a key marker of chondrogenic differentiation of FPSCs towards a “hyaline-like” phenotype. All scaffold variants demonstrated low levels of type I collagen deposition, a marker of fibro-cartilaginous differentiation and weak staining for type X collagen, a marker of hypertrophic differentiation. This collagen deposition profile is typical for FPSCs which generally undergo attenuated hypertrophic differentiation when compared to bone marrow derived MSCs ⁷². The correlation between increase sGAG production and decreased AC-ECM and glyoxal concentrations may be a result of a greater interconnectivity of internal pores, allowing for greater cell infiltration to the centre of the scaffold⁷³⁻⁷⁵. This hypothesis may explain the more robust, uniform sGAG staining observed in the 10mg/ml –

5mM scaffolds. Alteration of the freeze drying kinetics used or other methods to increase the pore size of these scaffold may further enhance the chondro-inductivity of the scaffolds.

The 10mg/5mM scaffold promoted the development of the stiffest cartilaginous tissue after 28 days in culture. This scaffold also supported the highest levels of sGAG deposition, with previous studies confirming a strong correlation between sGAG deposition and the equilibrium and dynamic moduli in FPSC-derived engineered cartilage tissues⁷⁰. Traditionally, AC-ECM scaffolds have been fabricated from particulated (usually cyromilled) AC, mixed with water to form a slurry and then freeze dried to form a scaffold. These scaffolds would generally maintain the majority of GAGs within the matrix, unlike the solubilized AC-ECM scaffolds we have developed here. Their overall porosity is typically lower than solubilised ECM scaffolds, although the mean pore size is typically higher. Previous work from our laboratory has explored the use of such particulated AC-ECM scaffolds for cartilage tissue engineering, using human FPSCs from the same donor, as well as the same seeding density, scaffold size and time-point (28 days), allowing us to compare the matrix deposited in both scaffold types. Both scaffold types supported the robust chondrogenic differentiation of FPSCs, with slightly higher levels of sGAG deposition within the particulated (cyromilled) scaffolds (120µg Vs. 115µg GAG per scaffold respectively) after 28 days in culture¹⁰. It should be noted, however, that the particulated scaffolds used in that study were not decellularized and lack the elastic properties of the solubilized and glyoxal cross-linked AC-ECM scaffolds (See Supplementary Fig. 2)

It should be noted that the mechanical properties of the generated cartilaginous tissues described in this study are significantly lower than the properties of native articular cartilage, which has a compressive modules of 0.1 to 2 MPa⁷⁶. Therefore, if the proposed scaffold is to be used for the *in vitro* engineering of functional cartilage grafts for subsequent implantation, further optimization of the culture conditions and/or the scaffold design may be required. In

conclusion, this study details the successful fabrication of porous articular cartilage ECM derived scaffolds, which were sufficiently decellularized for putative clinical use, did not elicit a potentially detrimental immunological response *in vitro* and were shown to promote robust chondrogenic differentiation of FPSCs. Due to the characteristics presented here, these scaffolds have the potential to be used in clinical applications, either in combination with current microfracture techniques, or seeded with cells prior to implantation, in order to treat focal cartilage defects.

Acknowledgements

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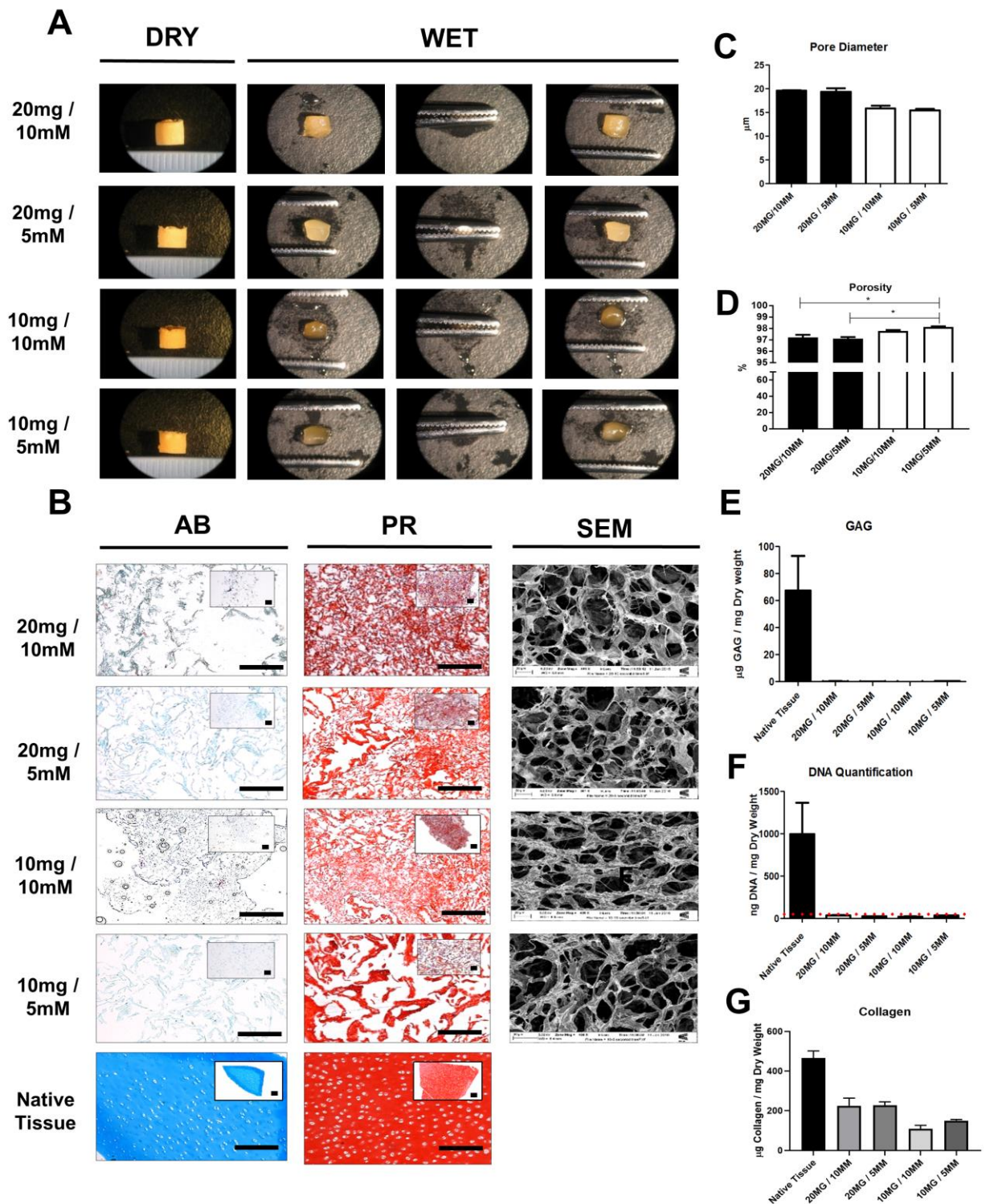


Figure 1: Characterisation of solubilised ECM derived scaffolds. Visualisation of the physical characteristics of the cell free ECM derived scaffolds was obtained by stereoscopic photography, histological staining and scanning electron microscopy (SEM) (A+B). Scaffolds contained either 10mg or 20mg of solubilised ECM, cross-linked with either 5mM or 10mM glyoxal. Mean pore diameter (C) and scaffold porosity (D) was obtained from 4 scaffolds per

group. sGAG (E),DNA (F) collagen (G) levels of scaffolds were quantified by biochemical assay (n=4). Dashed line in F represents 50ng/mg sufficient decellularization threshold. Scale bar =200µm in histological images and 20µm in SEM images. AB= Alcian Blue, PR= Picrosirius Red.* $P \leq 0.05$, ** $P \leq 0.01$ and *** $P \leq 0.001$.

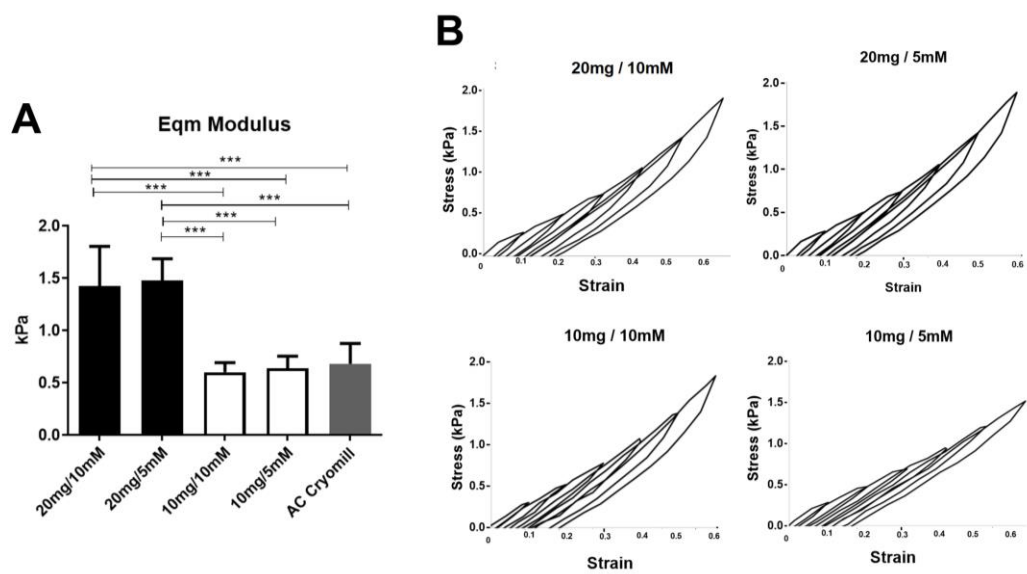


Figure 2: Mechanical properties of solubilised AC-ECM scaffolds. To obtain the cell free AC-ECM scaffold equilibrium modulus, stress relaxation tests were performed whereby 30% strain was applied to the samples (A). All scaffolds tested demonstrated elastic properties as visualised by the stress strain curves (B). *** $P \leq 0.001$, n=4. Scaffolds contained either 10mg or 20mg of solubilised ECM, cross-linked with either 5mM or 10mM glyoxal.

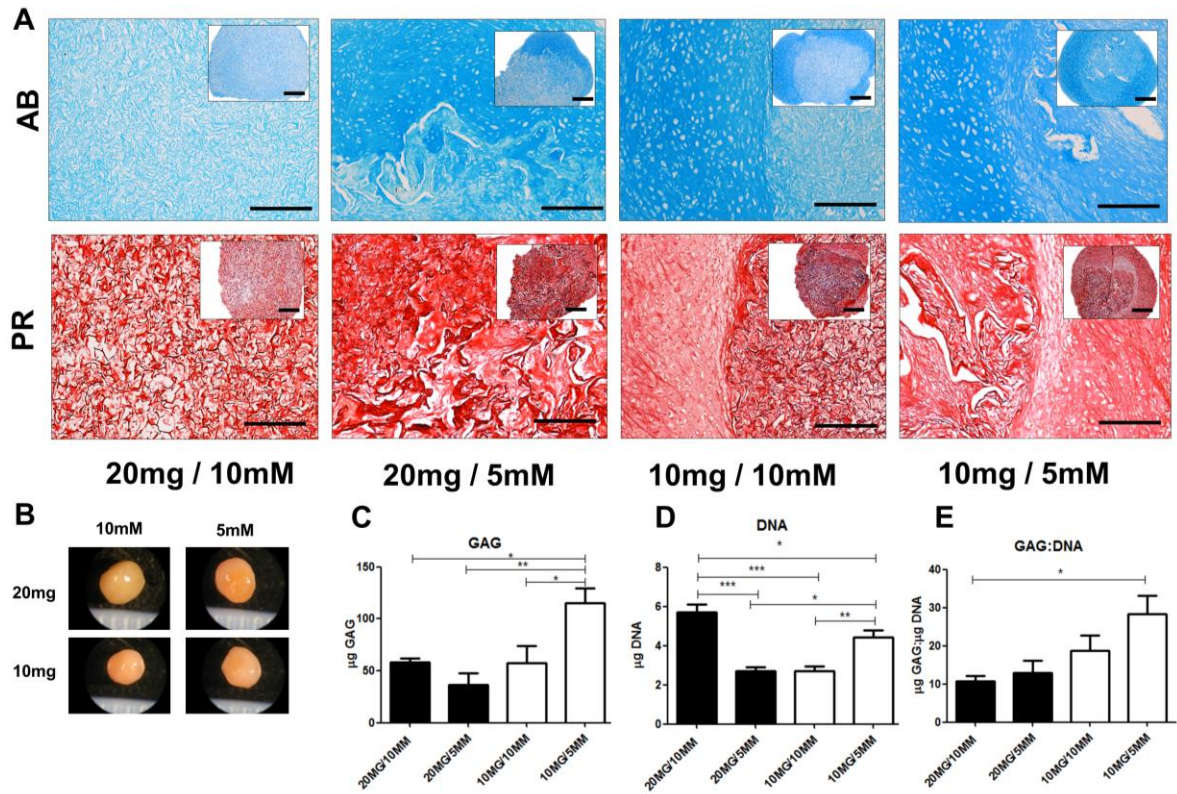


Figure 3: Robust chondrogenesis observed in 10mg / 5mM AC-ECM scaffolds after 28 days *in vitro*. sGAG and collagen was deposited by FPSCs in all scaffold groups (A). Stereoscopic micrographs show cartilage-like tissue formation in all groups, indentations on scale bar =1mm (B). Biochemical quantification reveals the greatest sGAG deposition in the 10mg / 5mM group (C-E). Scale bar =200 μm . AB= Alcian Blue, PR= Picrosirius Red. * $P \leq 0.05$, ** $P \leq 0.01$ and *** $P \leq 0.001$. Combined results from 3 individual donors, n=4 per donor.

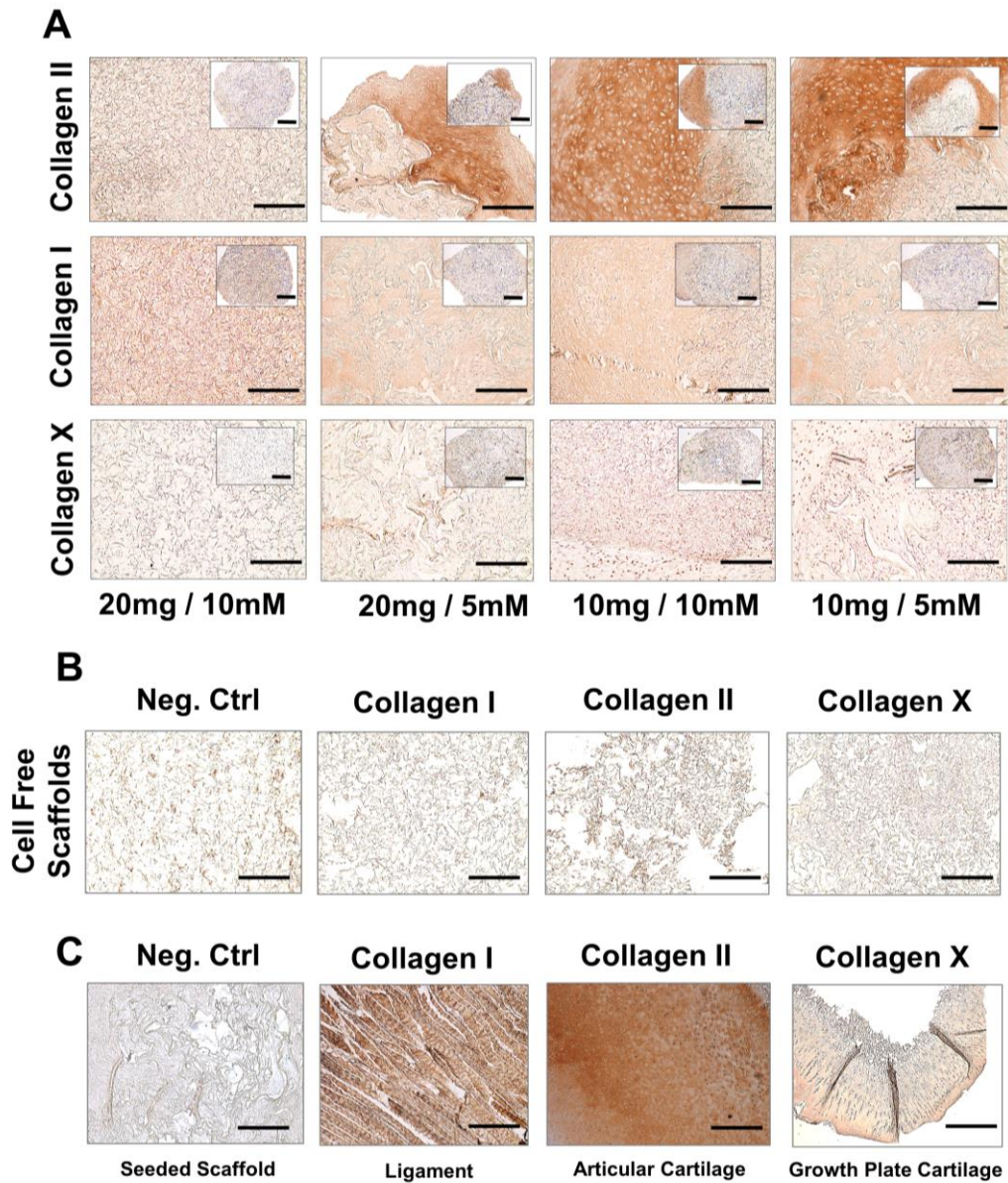


Figure 4: AC-ECM scaffolds seeded with FPSCs immunohistochemical analysis. Immunohistochemical analysis revealed that 20mg / 5mM, 10mg / 10mM and 10mg / 5mM groups all deposited type II collagen after 28 days. Also collagen deposition typical of differentiating FPSCs was observed, that is low deposition of type I collagen and type X collagen (A). Cell free scaffolds were subjected to the same immunohistochemistry procedure as in A to demonstrate no positive staining was present prior to cell seeding. Negative control images are without primary antibody only (B&C). Tissue specific positive controls for immunohistochemistry demonstrate positive staining for collagen types I, II and X in ligament, AC and growth plate cartilage respectively (C) Scale bar =200 μ m

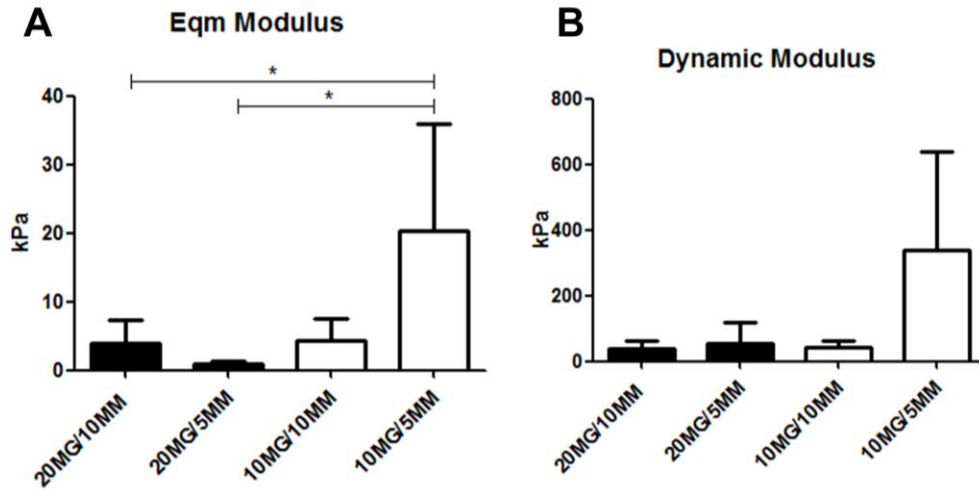


Figure 5: Mechanical properties of cell seeded AC-ECM scaffolds after 28 days *in vitro*. To obtain the cell seeded AC-ECM scaffold equilibrium modulus, stress relaxation tests were performed whereby 10% strain was applied to the samples (A). Dynamic modulus was obtained by applying a cyclic strain of 1% at 1 Hz to the scaffolds (B). Combined results from 3 individual donors, n=4 per donor. * $P \leq 0.05$.

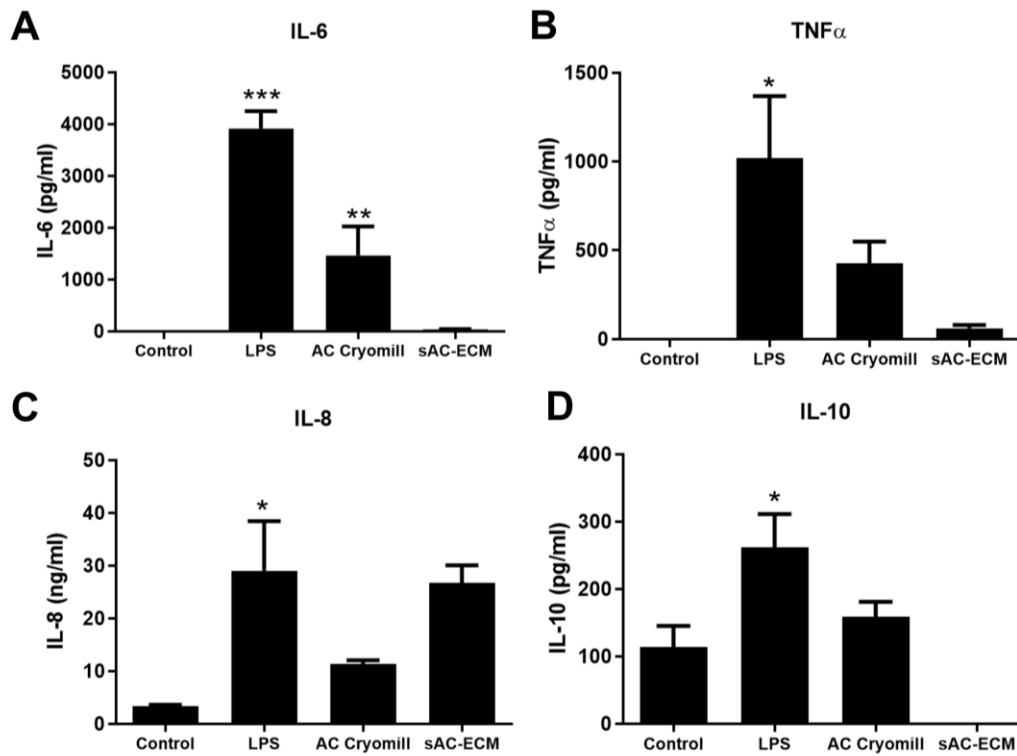
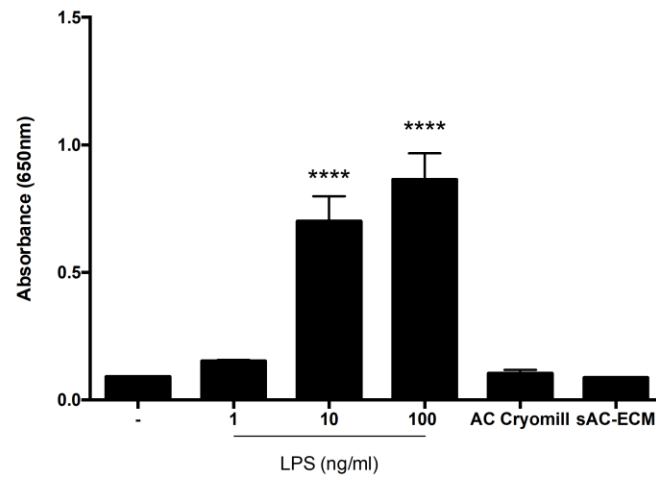
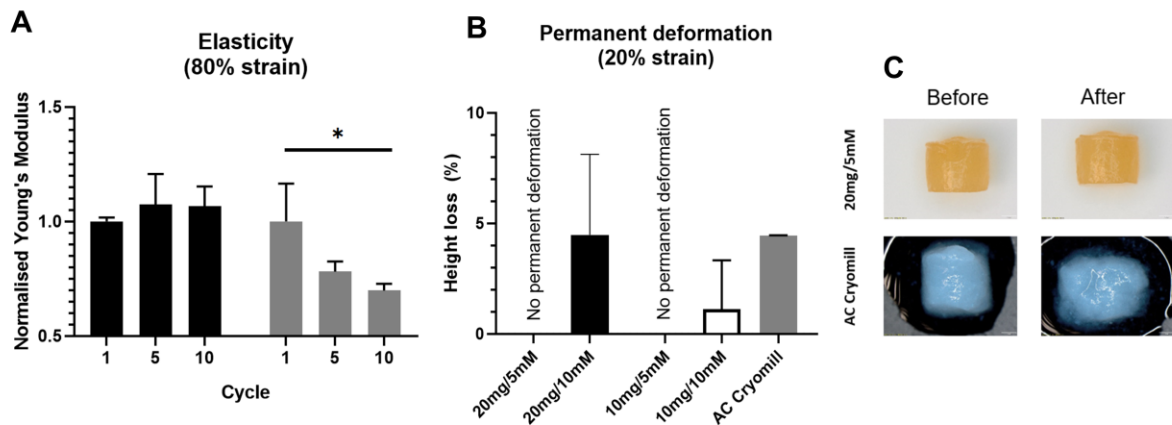


Figure 6: Effect of decellularisation of IL-6, TNF α , IL-8 and IL-10 cytokine production. Primary human macrophages (1×10^6 cells/ml) were treated with native or solubilised articular cartilage for 24 hours. As a positive control, macrophages were treated with 100 ng/ml LPS for 24 hours. (A) IL-6, (B) TNF α , (C) IL-8 and (D) IL-10 cytokine production was quantified in cell supernatants by ELISA. Unstimulated macrophages are the control group in all assays. Results shown are means (\pm SD) for triplicate cultures and are representative of 3 independent experiments.



Supplementary Figure 1: Confirmation of Endotoxin free ECM Scaffolds. ECM scaffold preparations were shown to be endotoxin free, using the HEK-Blue™ hTLR4 assay system (Invivogen). The expression of SEAP, as measured by absorbance at 650nm, by scaffold treated macrophages was comparable to untreated control cells. As a positive control, cells were treated with LPS (1 ng/ml, 10 ng/ml or 100 ng/ml).



Supplementary Figure 2: Mechanical testing to demonstrate the different behaviour of solubilised ECM scaffolds and those produced using particulated ECM (AC-Cryomill). Cumulative percentage of the total height loss and height loss per cycle was calculated to characterize the super-elastic properties of the constructs. The percentage height loss of the scaffolds was determined at 20% strain (A). Scaffolds (Solubilized and AC-cryomilled) were subjected to 80% strain to determine elasticity (B&C).

